

Tunnelling conduction process introduction during the degradation phenomena process in SnO₂ and ZnO varistors

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Abstract

An adjustment on the electrical degradation phenomena promoted by the application of low *dc* voltages on SnO₂ and ZnO based varistors is presented in this work. The electrical properties of varistors systems were highly damaged after treating the samples. Variations on the potential barrier height and width determined via impedance spectroscopy measurements are discussed on the basis of atomic defects at boundary regions and changes in the donor and surface states densities. Several authors consider thermionic conduction as the dominant process in SnO₂ and ZnO varistors but, for an appropriate barriers description, tunnelling current through the barriers must be considered.

The most studied varistor system, being currently used for commercial purpose, is based on a ZnO composition, as initially developed by Matsuoka *et al.* around 1970 [1]. Pianaro *et al.* [2] developed a SnO₂-based varistor composition, which has also been extensively studied and it is suggested to possess some advantages when compared to the ZnO-based compositions [3,4]. The purpose of this work is to gain knowledge in the degradation phenomena promoted by the application of a *dc* voltage and current pulses on SnO₂ and ZnO-based varistors. Variations on the potential barrier height and width determined via impedance spectroscopy analysis considering the thermionic and tunnelling conduction are discussed.

The ZnO-based varistor composition was 95.4% ZnO + 1.5% Sb₂O₃ + 1% NiO + 0.1% SiO₂ + 0.5%, whereas the SnO₂-based varistor composition was 98.9% SnO₂ + 1% CoO + 0.05% Nb₂O₅ + 0.05% Cr₂O₃. The high voltage circuit to generate the pulses of 8/20 μs was based on three capacitors having a total capacitance of 2.25 μF. Before and after the degradation process, each sample was electrically characterized by Impedance Spectroscopy (IS). In a first approach, a simple model considering the thermionic conduction was used. After that, experimental results were fitted using a sophisticated model which considers the contribution of the thermionic and tunnelling components to the electronic conduction. From Table 1, it can be seen that the barrier height is not greatly affected after degradation process even though an increase in *C_{gb}* was registered after each electrical stressing step.

Table 1: Grain boundary capacitance (*C_{gb}*) and resistance (*R_{gb}*) extracted from IS analysis, barrier height (*φ*) and donors concentration (*N_d*) obtained through different approaches to the electrical conduction.

| Sample | <i>R_{gb}</i> (Ω) | <i>C_{gb}</i> (F) | Thermionic conduction | | Thermionic plus tunnelling | | <i>J_{therm}</i> (A/cm ²) | <i>J_{tun}</i> (A/cm ²) | <i>J_{total}</i> (A/cm ²) |
|-------------------------------|---------------------------|---------------------------|---------------------------|---|----------------------------|---|---|---|---|
| | | | <i>φ_b</i> (eV) | <i>N_d</i> (m ⁻³) | <i>φ_b</i> (eV) | <i>N_d</i> (m ⁻³) | | | |
| ZnO before | 5,90x10 ⁶ | 3,82 x10 ⁻¹¹ | 0,778 | 4,19 x10 ²¹ | 0,810 | 5,410 x10 ²¹ | 6,5665 x10 ⁻⁸ | 5,157 x10 ⁻⁸ | 1,195 x10 ⁻⁷ |
| ZnO after c.c | 5,10 x10 ⁵ | 2,63 x10 ⁻¹⁰ | 0,716 | 1,84 x10 ²³ | 0,768 | 3,365 x10 ²³ | 3,3652 x10 ⁻⁷ | 2,176 x10 ⁻⁶ | 2,512 x10 ⁻⁶ |
| ZnO after-pulses | 5,70 x10 ³ | 1,44 x10 ⁻⁹ | 0,600 | 3,42 x10 ²⁴ | ----- | ----- | ----- | ----- | 2,234 x10 ⁻⁴ |
| SnO ₂ before | 3,77 x10 ⁶ | 1,44 x10 ⁻¹⁰ | 0,749 | 2,86 x10 ²³ | 0,781 | 7,417 x10 ²² | 2,0229 x10 ⁻⁷ | 4,999 x10 ⁻⁷ | 7,028 x10 ⁻⁷ |
| SnO ₂ after c.c | 2,88 x10 ⁶ | 1,620 x10 ⁻¹⁰ | 0,742 | 3,58 x10 ²³ | 0,776 | 9,327 x10 ²² | 2,465 x10 ⁻⁷ | 7,064 x10 ⁻⁷ | 9,528 x10 ⁻⁷ |
| SnO ₂ after pulses | 1,76 x10 ⁶ | 3,46 x10 ⁻⁹ | 0,582 | 6,81 x10 ²⁵ | 0,595 | 3,956 x10 ²¹ | 2,8196 x10 ⁻⁴ | 1,814 x10 ⁻⁴ | 4,633 x10 ⁻⁴ |

References

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